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Reversible Switching between Nonporous and Porous Phases of a New SIFSIX Coordination Network Induced by a Flexible Linker Ligand

Bai-Qiao Song, Qing-Yuan Yang, Shi-Qiang Wang, Matthias Vandichel, Amrit Kumar, Clare Crowley, Naveen Kumar, Cheng-Hua Deng, Victoria GasconPerez, Matteo Lusi, Hui Wu, Wei Zhou, and Michael J. Zaworotko*



ABSTRACT: Closed-to-open structural transformations in flexible coordination networks are of potential utility in gas storage and separation. Herein, we report the first example of a flexible $\text{SiF}_6^{2^-}$ -pillared square grid material, $[\text{Cu}(\text{SiF}_6)(\text{L})_2]_n$ (L = 1,4-bis(1-imidazolyl)benzene), **SIFSIX-23-Cu**. **SIFSIX-23-Cu** exhibits reversible switching between nonporous (β 1) and several porous (α , γ 1, γ 2, and γ 3) phases triggered by exposure to N₂, CO₂, or H₂O. In addition, heating β 1 to 433 K resulted in irreversible transformation to a closed polymorph, β 2. Single-crystal X-ray diffraction studies revealed that the phase transformations are enabled by rotation and geometrical contortion of L. Density functional theory calculations indicated that L exhibits a low barrier to rotation (as low as 8 kJmol⁻¹) and a rather flat energy surface. In situ neutron powder diffraction studies provided further insight into these sorbate-induced phase changes. **SIFSIX-23-Cu** combines stability in water for over a year, high CO₂ uptake (ca. 216 cm³/g at 195 K), and good thermal stability.

Metal-organic materials (MOMs)¹ such as metal-organic frameworks (MOFs)² and porous coordination polymers $(PCPs)^3$ offer potential solutions to the high-energy footprint, costs, and risks associated with storage or purification of gases and vapors.⁴ Whereas rigid microporous MOMs typically display type I isotherms, flexible MOMs (FMOMs)⁵ can exhibit stepped isotherms from structural contraction (expansion) under reduced (increased) pressure, i.e., breathing or swelling.⁶ FMOMs that exhibit stepped type F-IV isotherms, sudden switching from a nonporous (closed) activated phase to a porous (open) phase, are of particular interest for pressure swing adsorption (PSA) gas storage,⁷ as negligible adsorbate is present at low pressure.⁸ Closed-toopen transformations might also facilitate gas separation if adsorbates selectively induce transformations.⁹ Thus far, only ca. 150 FMOMs of the >20 000 porous MOMs reported are known to be flexible,^{5b,10} and just a handful exhibit type F-IV isotherms with saturated uptake of >200 cm³/g.¹¹ None of these high-uptake FMOMs are sustained by inorganic linker ligands, a matter we address herein.

The "pillared sheet" platform of materials comprised of cationic square grid lattice (sql) sheets and hexafluorosilicate $(SiF_6^{2-}, "SIFSIX")$ pillars¹² exhibit benchmark performance with respect to separation of industrial gases such as CO₂, SO₂, and small-molecule hydrocarbons.¹³ Whereas modeling studies indicate that rotation of pyrazine rings in SIFSIX-3-M (M = Ni, Fe) can cause inflections in Xe adsorption isotherms, the framework remains rigid.¹⁴ SIFSIX nets can also transform into nonporous sql or sql-c* nets under humid conditions.¹⁵ Herein, we introduce the first flexible SIFSIX net, [Cu(SiF₆)-

 $(L)_2]_n$ (L = 1,4-bis(1-imidazolyl)benzene), SIFSIX-23-Cu (Figure 1).

Layering 1:1 MeOH/H₂O onto CuSiF₆ in H₂O followed by further layering of a solution of L in MeOH afforded purple, needle-shaped crystals of SIFSIX-23-Cu·xMeOH·yH₂O (see



Figure 1. (a) Self-assembly strategy used to design SIFSIX-23-Cu. (b) 1D channels along the *a*-axis in the as-synthesized form, SIFSIX-23-Cu- α .

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Supporting Information (SI)), SIFSIX-23-Cu- α . Cu(II) cations are octahedrally coordinated to four equatorial L ligands (two L(*syn*) and two L(*anti*)), while SIFSIX anions occupy the axial sites to afford a noninterpenetrated **pcu** topology framework (Figure 1a). The CuL₂ sql net undulates due to out-of-plane protrusion of "V"-shaped L(*syn*) linker ligands. The pillars adopt a *cis*-bridging mode, unlike other coordination networks involving SIFSIX anions (Figures S1 and S2).^{12,13,16} SIFSIX-23-Cu- α possesses a 1D channel of diameter 5.4 × 4.0 Å² along the *a*-axis (Figure 1b) with a calculated accessible void volume of 41.8% (Figure S3).

We found that desolvation induces SIFSIX-23-Cu- α to undergo multiple phase transformations to SIFSIX-23-Cu- γ 1, - γ 2, - γ 3, and a solvent-free closed form, SIFSIX-23-Cu- β 1 (Figures 2a and S4, Table S1). Whereas the space group and



Figure 2. (a) Guest-induced reversible structural transformations occur between the α , $\gamma 1$, $\gamma 2$, $\gamma 3$, and $\beta 1$ phases. (b) A parallelogram defines the structural parameters of each phase. Deformation of the parallelogram during the α to $\gamma 3$ (c) and the $\gamma 3$ to $\beta 1$ (d), transformations. (e) Overlapping parallelograms of the α and $\beta 1$ phases highlight structural distortions.

unit cell parameters vary, the connectivity of the SIFSIX-23-Cu network is preserved even with extreme deformation (Figure 2b, Table S2). The α to γ 3 transformation caused the parallelogram of the sql net to undergo hinge-like motion with negligible change in edge length ($\Delta d_{max} = 0.2$ Å) but a decrease in \angle Cu-Cu-Cu from 90° (α) to 53.4° (γ 3) (Figure 2b and c). Concomitantly, one diagonal of the parallelogram reduced from 18.46 Å (α) to 11.67 Å (γ 3), whereas the other expanded from 18.46 Å (α) to 23.19 Å (γ 3). An overall reduction in guest-accessible volume from 41.8% (α) to 20.4% (γ 3) was calculated by PLATON.¹⁷ Removal of solvent from γ 3 afforded SIFSIX-23-Cu- β 1, a nonporous phase (Figures 2b and S5 and S6) with 34.5% volume reduction vs the α phase. The transformation from γ 3 to β 1 differs from the other phase changes (Figure 2b and d). FMOMs rarely exhibit such extreme structural changes.¹¹ β 1 reverted to α after immersion in MeOH, EtOH, or CH₃CN (Figure S7).

Structural analysis of the SIFSIX-23-Cu phases (Figures 2e and S8-S11, Table S2 and S3)¹⁸ reveals hinge-like motion in the α to $\gamma 1$ transformation that originates from rotation of imidazolyl rings and reorientation of phenyl rings. Indeed, rotation of L is a feature of each transformation (Figure S8). In the $\gamma 1$ to $\beta 1$ transformation, the coordinated N atom acts as a hinge or "kneecap" at which the Cu-imidazolyl junction bends (Figure S9). For the $\gamma 3$ to $\beta 1$ transformation, one imidazolyl ring of L(syn) bends at the $N(1)_{imidazolyl}$ atom to subtend an angle of 16.9° between the $C_{phenyl}-N_{imidazolyl}$ bond and the imidazolyl plane (Figure S10), ^{8d}, ^{11d}, ¹⁹ enabling shrinkage of the Cu-L(syn)-Cu edge in β 1. These structural changes are mainly associated with L(syn) (Figure S11). C-H…F hydrogen bonds between SIFSIX anions and CH moieties of aromatic groups (Table S4) are often present in SIFSIX coordination networks (Figure S12).¹³ Density functional theory (DFT) calculations indicated that L exhibits a low rotational barrier (<8 kJ/mol) and a rather flat energy surface with minima corresponding to L(syn) and L(anti). A Cambridge Structural Database (CSD)²⁰ survey revealed that dihedral angles between phenyl and imidazolyl rings are consistent with those in L(syn) and L(anti) (Figures S13-S15). The energy difference of L(syn) calculated before and after bending from γ 3 to β 1 is also small (<16 kJ/mol) (Figure S16). The nature of L helps to explain the "softness" of SIFSIX-23-Cu.²¹

Variable-temperature PXRD studies of the α to $\beta 1$ transformation (Figures 3a and S17) revealed a new phase at



Figure 3. (a) VT-PXRD of SIFSIX-23-Cu from 295 to 515 K. (b) TGA and DSC of SIFSIX-23-Cu. (c) Structural changes in the β 1 to β 2 transformation. (d) Relative energies of the structurally characterized phases (kJ/mol, PBE and PBE-dDsC).

425 K as also suggested by DSC (Figure 3b). That no weight loss was seen in the corresponding TGA curve indicates that this was a temperature-controlled rather than guest-controlled transformation. Heating $\beta 1$ or any of the α phases at 433 K afforded single crystals of SIFSIX-23-Cu- $\beta 2$. Single-crystal Xray diffraction (SCXRD) characterization revealed that $\beta 2$ exhibits a slightly smaller cell volume than that of $\beta 1$. The connectivity is unchanged if *anti*-to-*syn* switching of L(*anti*) and swing motion of SIFSIX anions (Figure 3c) had occurred. That $\beta 2$ is more stable than $\beta 1$ is supported by its higher density, structural parameters which suggest less strain (Figure S11) and lack of reversibility after immersion in water, MeOH, EtOH, or CH₃CN (Figure S18) and periodic DFT calculations (PBE and PBE-dDsC). The DFT calculations revealed that small global energy differences between the α to $\gamma 3$ phases (Figure 3d) are consistent with the reversibility of the structural transformations (Table S5, Figure S19). The energy difference between $\beta 1$ and $\gamma 3$ is 86.7 kJ/mol, a value that can be offset by adsorption. The $\beta 2$ phase was determined to be relatively more stable (-222 kJ/mol vs α).

Since the closed-to-open transformation from $\beta 1$ to α was induced by solvent, we anticipated that gases might also trigger switching. The CO₂ isotherm of $\beta 1$ at 195 K indeed displayed switching with two steps (Figure 4a). The first step at ca. 2.4 mmHg (adsorbed CO₂ = 84 cm³/g) gave a micropore volume of 0.16 cm³/g, close to that estimated from the crystal structure of $\gamma 3$ (0.15 cm³/g). The second step occurred after saturation (ca. 216 cm³/g) with a micropore volume of 0.40 cc/g, in good agreement with the structure of α (0.40 cm³/g). The Langmuir surface areas for the phases in the first and second steps



Figure 4. (a) Gas sorption isotherms of $\beta 1$ at 195 K (CO₂) and 77 K (N₂). (b) CO₂ sorption isotherms of $\beta 1$ at 273 and 298 K. (c) Adsorption/desorption cycles of $\beta 1$ at 273 K. (d) High-pressure CO₂ sorption isotherms at 25 to 55 °C. (e) In situ NPD patterns at various conditions. At 77 K, 1 bar N₂ trace $\beta 1$ is present because of large sample size (>1 g) and slow kinetics; equilibrium took >24 h. (f) Water sorption isotherm of $\beta 1$ at 298 K.

calculated from the CO₂ isotherm at 195 K are 451 and 941 m² g⁻¹, respectively (Figures S20 and S21). β 1 also displayed gate opening for N₂ at 77 K with an onset pressure of 142 mmHg and saturated uptake of 160 cm³/g at 738 mmHg (Figure 4a). In contrast, the N₂ (77 K) and CO₂ (195 K) isotherms for β 2 revealed no uptake (Figure S22).

In situ neutron powder diffraction (NPD) data were collected under various conditions to further study these switching events (Figure 4e). The diffraction pattern obtained at 0 bar matched that calculated for β 1. For P = 0.02 bar at 195 K (the first step), the NPD pattern resembled that calculated for γ 3. For P = 0.2 bar (the second step), the NPD pattern matched that calculated for α . The α phase reverted to β 1 at P = 0 bar and 298 K. NPD data of samples collected at 77 K under 1 bar N₂ validated the formation of α . The saturated uptake of N₂ was lower than that calculated from the pore volume based on the CO₂ sorption data (258 cm³/g), perhaps because at 77 K sorbate—sorbent interactions are strong enough to hinder diffusion.²² This guest-dependent switching behavior suggests that structural transformations are driven by host—guest interactions.

The low-pressure CO₂ sorption isotherms at near ambient temperature revealed no switching behavior induced by CO₂ at 298 K below 1 bar (Figures 4b and S23). However, $\beta 1$ exhibited a single step type F-IV isotherm at 273 K with a gateopening pressure of 505 mmHg and saturated uptake of 77 cm^3/g at 759 mmHg. The corresponding desorption isotherm displayed hysteresis. The calculated micropore volume of 0.15 cm^3/g agrees with that calculated from the crystal structure of γ 3 (0.15 cm³/g). In situ NPD data collected at 273 K and 1 bar CO₂ showed a similar NPD pattern to that obtained at 195 K for 0.02 bar CO₂, i.e., the γ 3 phase (Figure 4e). That this phase switching is reversible was demonstrated by 41 cycles of CO₂ adsorption/desorption (Figures 4c and S24–S26). PXRD confirmed structural integrity (Figure S27). CO₂ triggered closed to open switching with an onset gate-opening pressure of <1 bar at 273 K observed in the ELM family of materials and zeolitic imidazolate frameworks (ZIFs).^{8b,2}

High-pressure CO_2 sorption on $\beta 1$ (Figure 4d) resembled the isotherm obtained at 195 K. The onset gate-opening pressures for the first and second step at 298 K are around 2 and 5.8 bar, respectively. Interestingly, even at 55 °C CO_2 triggers gate-opening with a threshold pressure of *ca*. 5.3 bar, a pressure suitable for high-temperature gas sorption and separation. CO_2 -induced closed-to-open switching from $\beta 1$ to α at high pressure was found to be reversible (Figures S28– S30).

SIFSIX-23-Cu- β 1 also underwent closed-to-open switching induced by water vapor at 298 K (Figure 4f). The steep water uptake (2.8%) at low humidity matches the SCXRD results, which indicate that β 1 can capture one water molecule without a structural change. β 1 sequentially formed γ 3, γ 2, and γ 1 under atmospheric humidity (Figures S31 and S32) whereas water immersion caused β 1 to revert to α (Figures S7). SIFSIX-23-Cu- α retained single-crystallinity after immersion in water for one year (Figures 5 and S33 and S34). This strong hydrolytic stability contrasts with that of other SIFSIX materials.¹⁵

In summary, the new pillared square grid coordination network **SIFSIX-23-Cu** exhibits an uncommon *cis*-bridging coordination mode of SIFSIX anions and is to our knowledge the first hybrid coordination network that undergoes reversible guest-induced closed-to-open transformations. It is also one of



Figure 5. (a) As-synthesized crystals compared to those soaked in water for one year. (b) PXRD pattern of **SIFSIX-3-Cu** soaked in water for one year compared with that calculated for α . (b) The 195 K CO₂ and 77 K N₂ adsorption isotherms of samples before and after being soaked in water for one year.

the highest uptake materials with a type F-IV isotherm. We attribute these facile and reversible transformations to the low rotational barrier of L and the correspondingly flat energy surface.

ASSOCIATED CONTENT

③ Supporting Information

The Supporting Information is available free of charge at https://pubs.acs.org/doi/10.1021/jacs.0c01314.

Materials and methods, supporting figures, supporting tables, and supporting references (PDF) Crystallographic information file (CIF)

(CIF)

AUTHOR INFORMATION

Corresponding Author

Michael J. Zaworotko – Department of Chemical Sciences and Bernal Institute, University of Limerick, Limerick V94 T9PX, Republic of Ireland; orcid.org/0000-0002-1360-540X; Email: Michael.Zaworotko@ul.ie

Authors

- **Bai-Qiao Song** Department of Chemical Sciences and Bernal Institute, University of Limerick, Limerick V94 T9PX, Republic of Ireland
- Qing-Yuan Yang School of Chemical Engineering and Technology, Xi'an Jiaotong University, Xi'an 710049, China; orcid.org/0000-0002-1742-2088
- Shi-Qiang Wang Department of Chemical Sciences and Bernal Institute, University of Limerick, Limerick V94 T9PX, Republic of Ireland; orcid.org/0000-0003-1213-8317

- pubs.acs.org/JACS
- Matthias Vandichel Department of Chemical Sciences and Bernal Institute, University of Limerick, Limerick V94 T9PX,

Communication

- Republic of Ireland Amrit Kumar – Department of Chemical Sciences and Bernal
- Institute, University of Limerick, Limerick V94 T9PX, Republic of Ireland
- **Clare Crowley** Department of Chemical Sciences and Bernal Institute, University of Limerick, Limerick V94 T9PX, Republic of Ireland
- **Naveen Kumar** Department of Chemical Sciences and Bernal Institute, University of Limerick, Limerick V94 T9PX, Republic of Ireland
- **Cheng-Hua Deng** Department of Chemical Sciences and Bernal Institute, University of Limerick, Limerick V94 T9PX, Republic of Ireland
- **Victoria GasconPerez** Department of Chemical Sciences and Bernal Institute, University of Limerick, Limerick V94 T9PX, Republic of Ireland
- **Matteo Lusi** Department of Chemical Sciences and Bernal Institute, University of Limerick, Limerick V94 T9PX, Republic of Ireland
- **Hui Wu** NIST Center for Neutron Research, National Institute of Standards and Technology, Gaithersburg, Maryland 20899-6102, United States; © orcid.org/0000-0003-0296-5204
- Wei Zhou NIST Center for Neutron Research, National Institute of Standards and Technology, Gaithersburg, Maryland 20899-6102, United States; orcid.org/0000-0002-5461-3617

Complete contact information is available at: https://pubs.acs.org/10.1021/jacs.0c01314

Notes

The authors declare no competing financial interest.

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